



Mechanical and thermoelectric properties of $\text{Ca}_3\text{Co}_4\text{O}_9$ ceramics consolidated by SPS process

**D. Kenfoui¹, G. Bonnefont², D. Chateigner¹, G. Fantozzi²
M. Gomina¹ and J.G. Noudem¹**

¹Laboratoire CRISMAT, UMR 6508 CNRS / ENSICAEN
6, Boulevard Maréchal Juin, 14050 Caen Cedex 4
<http://www-crismat.ensicaen.fr>

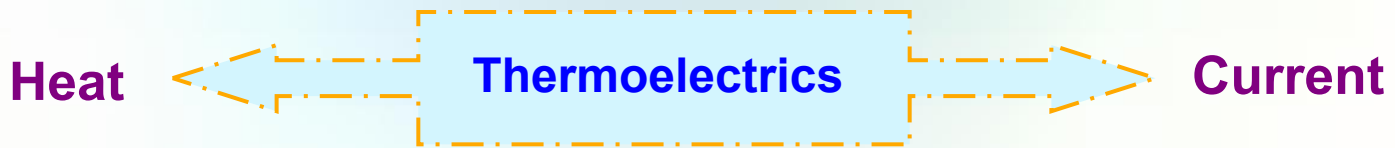
²Laboratoire MATEIS- INSA, Bât. B. Pascal, 5^o étage
7 avenue Jean Capelle 69621, Villeurbanne Cedex
<http://mateis.insa-lyon.fr>

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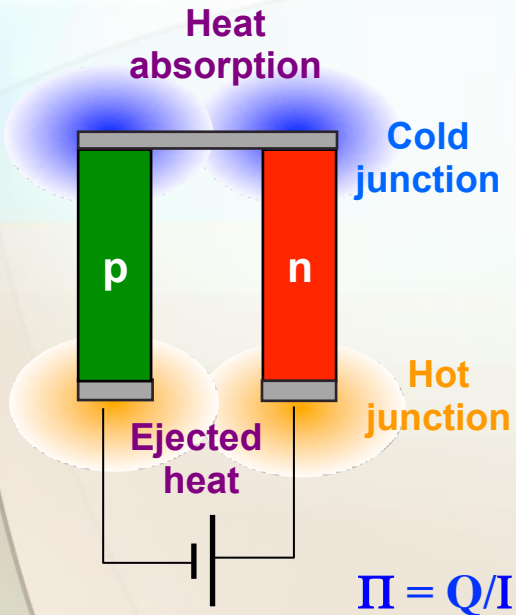
Outline

- **Thermoelectric Materials**
 - **Ca₃Co₄O₉ Oxides**
 - **Material preparation**
- **Microstructure and texture analyses**
- **Thermoelectric characterizations**
 - **Electrical resistivity and Seebeck coefficient**
 - **Power factor**
- **Mechanical properties**
 - **Young's modulus and Hardness**
 - **Strength**
- **Conclusions and outlooks**

Thermoelectric effects

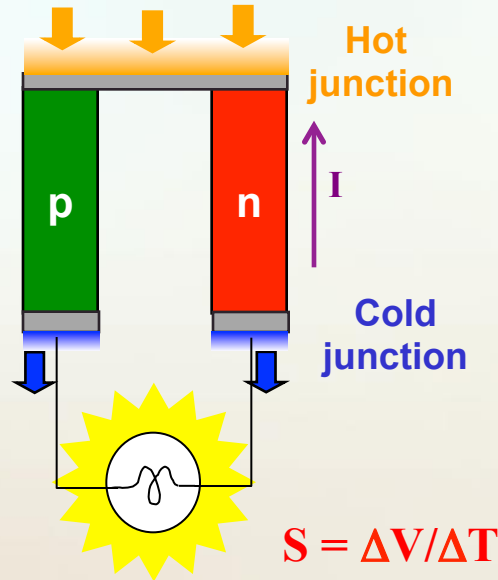


Peltier effect



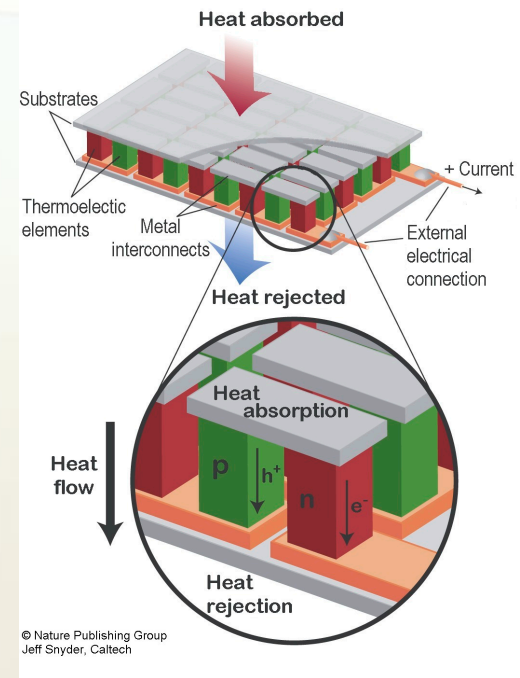
Refrigeration

Seebeck effect



Power generation

TE Modules



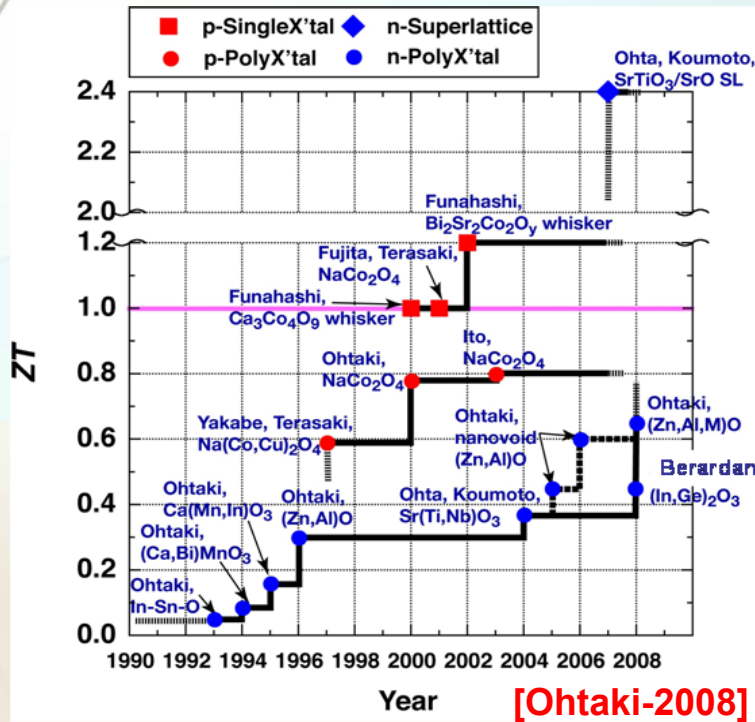
Applications : RTG* into space probes
(Galileo and Ulyssé)

Applications : Refrigerators (electronic systems)

* Radio-isotope Thermoelectric Generators

Thermoelectric Materials

TE oxides efficiency, ZT



→ The figure-of-merite: ZT

$$ZT = \frac{S^2}{\rho \cdot \kappa} T$$

[F.J. DiSalvo-1999]

S: Seebeck coefficient (V.K⁻¹)

T: Temperature (K)

ρ: Electrical resistivity (Ω.m)

κ: Thermal conductivity (Ω.m⁻¹.K⁻¹)

$$\kappa = K_e + K_{ph} \begin{cases} K_e: \text{electronic contribution} \\ K_{ph}: \text{phonon contribution} \end{cases}$$

Why the Ca₃Co₄O₉ oxydes (349) ?

- ✓ High thermal and chemical stabilities
- ✓ High oxidation resistance
- ✓ No toxic elements

Goal

❑ Master the 349 ceramics **microstructure** through **Spark Plasma Sintering - SPS** process

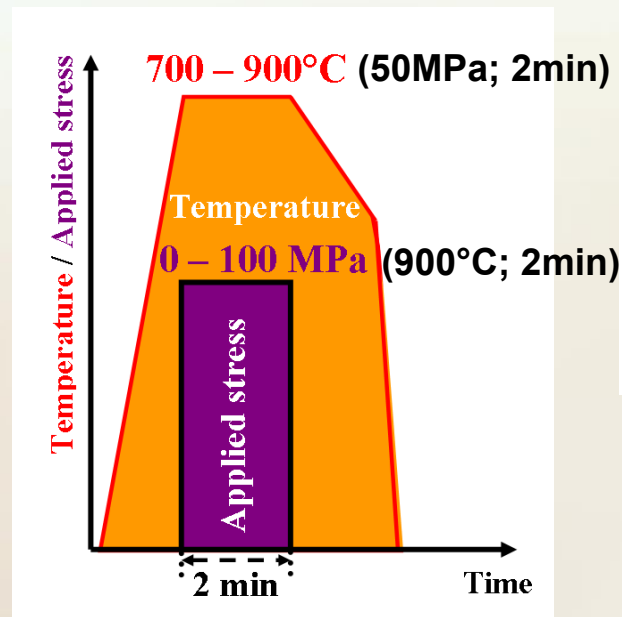
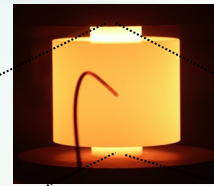
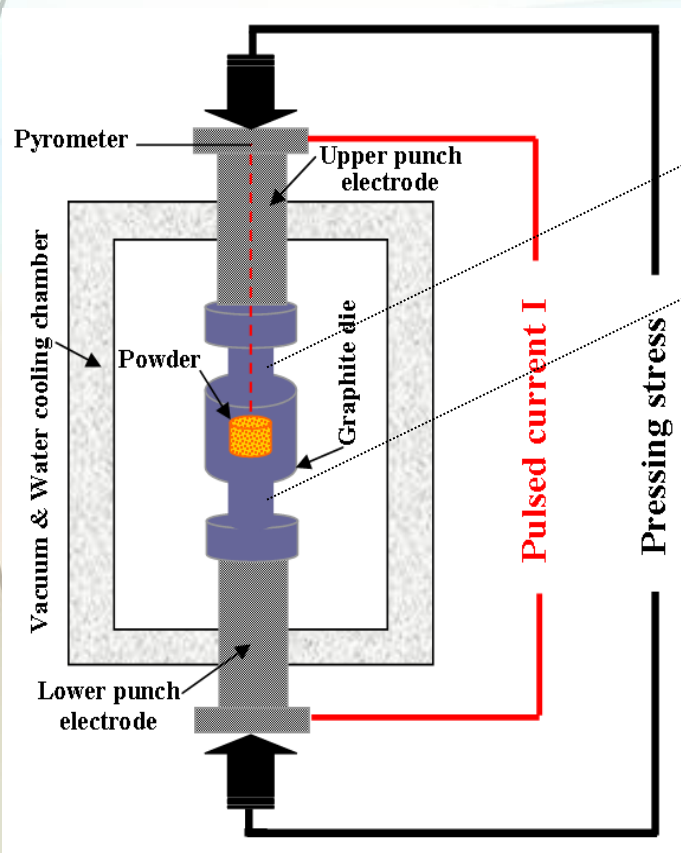
➡ Improve the **mechanical** and **thermoelectric properties** and correlate with the microstructure

➡ Integrate the **performing 349 ceramics** in **thermoelectric modules**

Elaboration of $\text{Ca}_3\text{Co}_4\text{O}_9$ ceramics

□ $\text{Ca}_3\text{Co}_4\text{O}_9$ synthesised by solid-state method from CaCO_3 and Co_3O_4 powders

□ Spark Plasma Sintering (SPS): new processing technique (since 2000)



SPS System / FCT HP D25/I

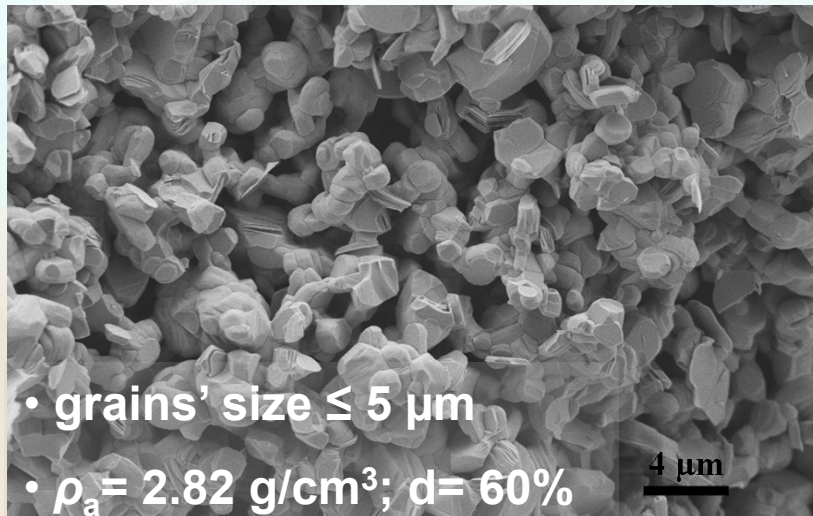
MATEIS- INSA, Lyon, FRANCE

[M. Tokita-1999]

Microstructure and Texture

■ Pressing stress effects (0 - 100 MPa)

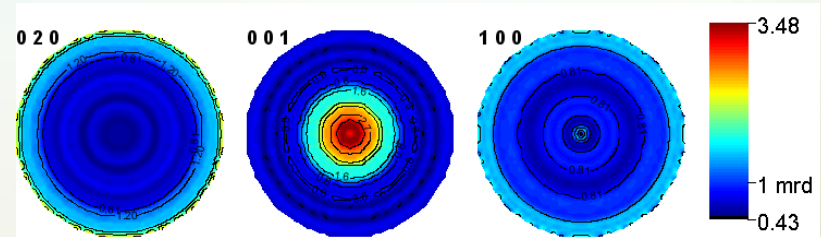
Conventional sintering (0 MPa; 920°C; 24 h)



- Grains randomly oriented
- High porosity → Weak density

(Theoretical density : 4.68 g/cm^3) [Masset - 2000]

➤ Pole Figures



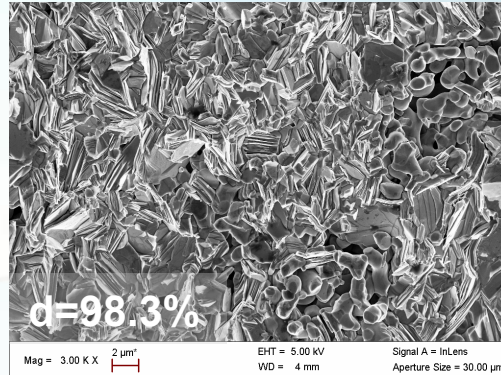
- The maximum of the **{001}** poles is **3.5 mrd**. (multiples of a random distribution)

➡ This sample does not exhibit texture

Microstructure and Texture

Pressing stress
30 MPa

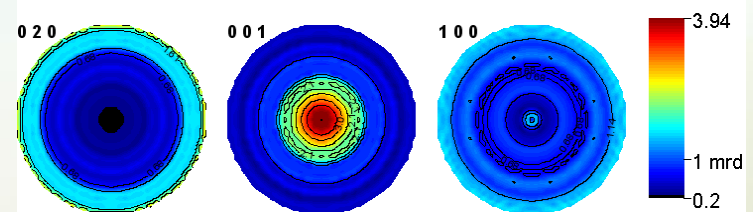
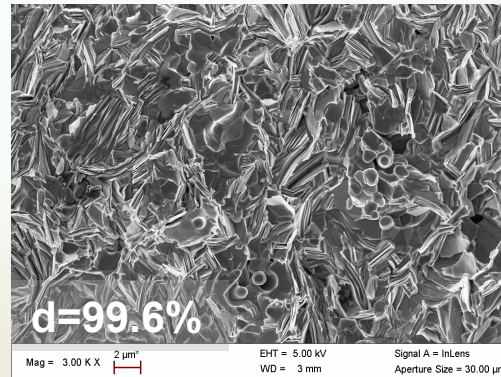
(900°C;2min)



- The maximum of the **{001}** poles is **3.74 mrd**.

50 MPa

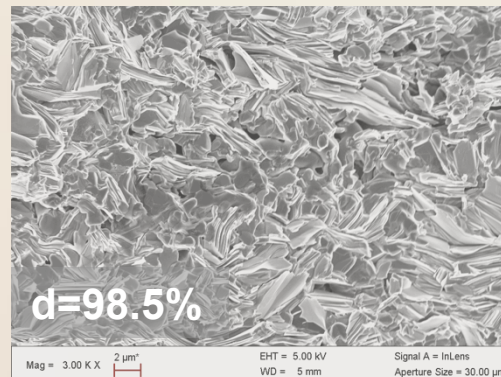
(900°C;2min)



- The maximum of the **{001}** poles is **3.94 mrd**.

75 MPa

(900°C;2min)

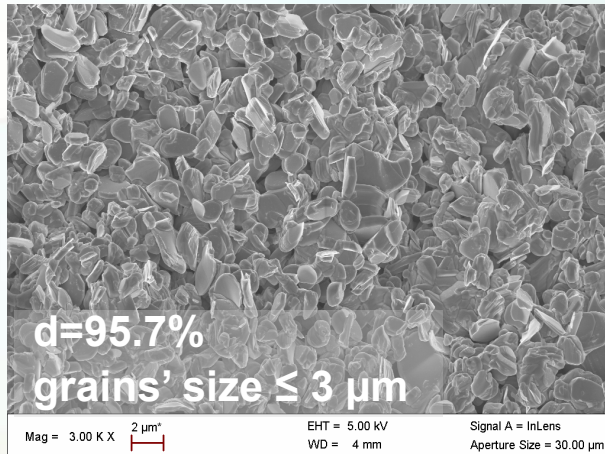


- The maximum of the **{001}** poles is **4.05 mrd**.

Microstructure and Texture

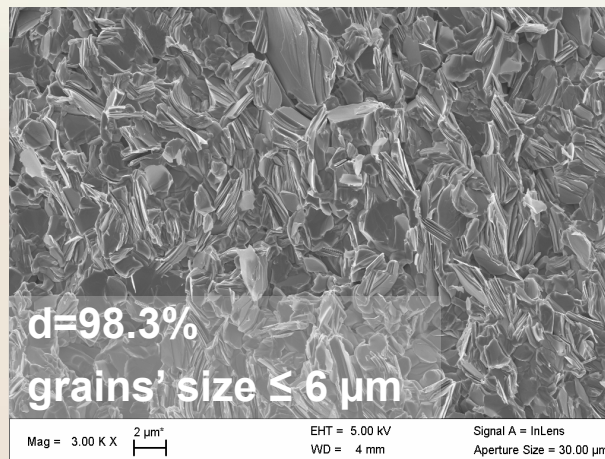
■ Temperature effects (700 – 900°C)

SPS
Temperature
700°C
(50 MPa; 2 min)



■ The maximum of the **{001}** poles is **2.29 mrd.**

850°C
(50 MPa; 2 min)



■ The maximum of the **{001}** poles is **4.41 mrd.**

Microstructure and Texture

- Bulk density and maximum of {001} poles values for different applied stress levels and temperatures

SPS stress (MPa)	0	30	50	75	100
Bulk density (%)	60	98.3	99.6	98.5	99.4
Maximum of {001} poles (m.r.d)	3.5	3.74	3.94	4.05	3.49

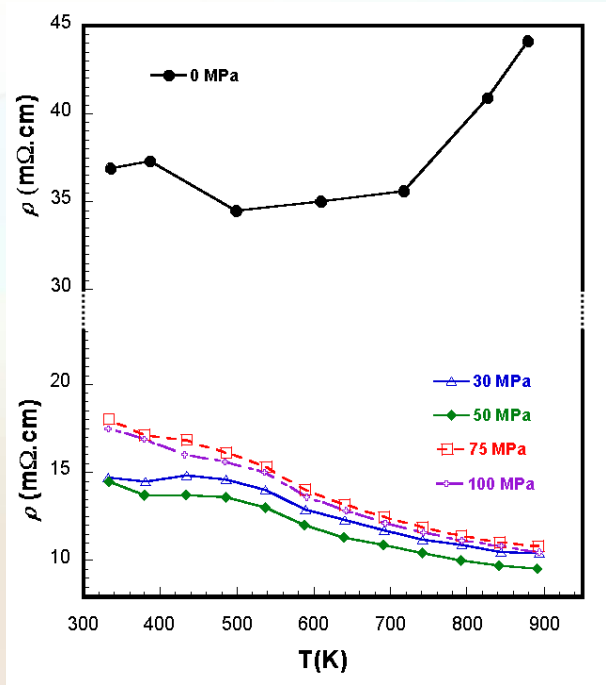
SPS temperature (°C)	700	750	800	850	900
Bulk density (%)	95.7	96.2	97.2	98.3	99.6
Maximum of {001} poles (m.r.d)	2.29	2.4	3.14	3.41	3.94

➔ SPS pressing stress and temperature : 50 MPa; 900°C

Thermoelectric characterization

■ Pressing stress effects (0 - 100 MPa)

➤ Resistivity and Seebeck coefficient measurements



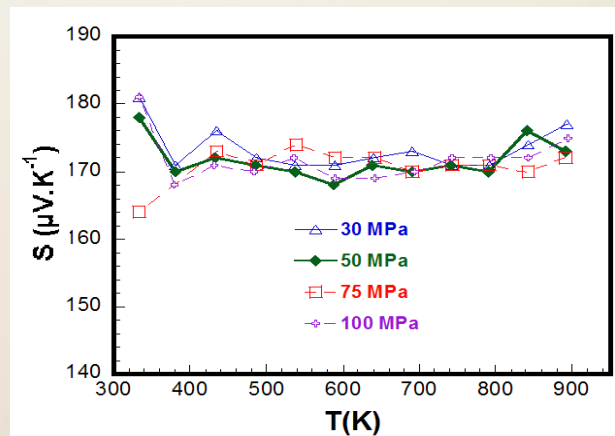
□ Seebeck coefficient is independent of the pressing stress levels (as it is expected for)

□ ρ reduces with the pressing stress up to 50 MPa

- due to the increase in the bulk density.

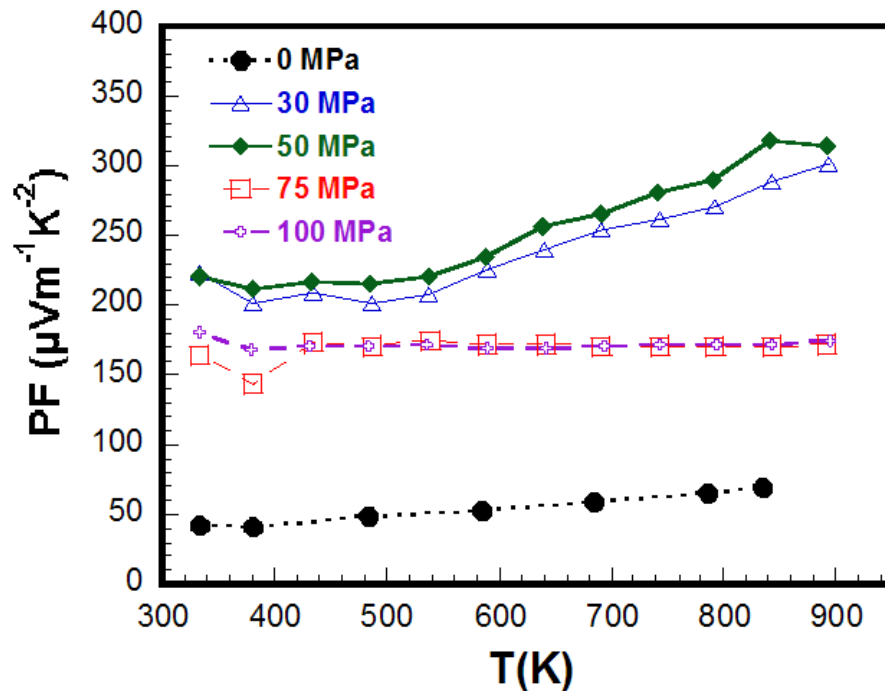
□ ρ increases for stress superior to 50 MPa

- due to the 'reappearance' of the weak zones at higher stress.



Thermoelectric characterization

➤ Power factor, $PF = \frac{S^2}{\rho}$



□ PF improves as the applied stress is increased up to **50 MPa**

□ Better PF values are obtained for SPS stress of **50 MPa**

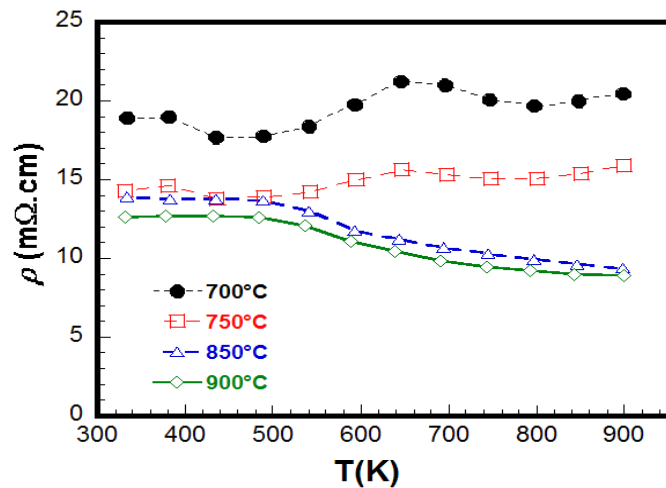
[Y. Zhou et al – 2003]
[Y. Liu et al. 2005]

➔ Better TE properties are found for 50 MPa

Thermoelectric characterization

■ Temperature effect (700 – 900°C)

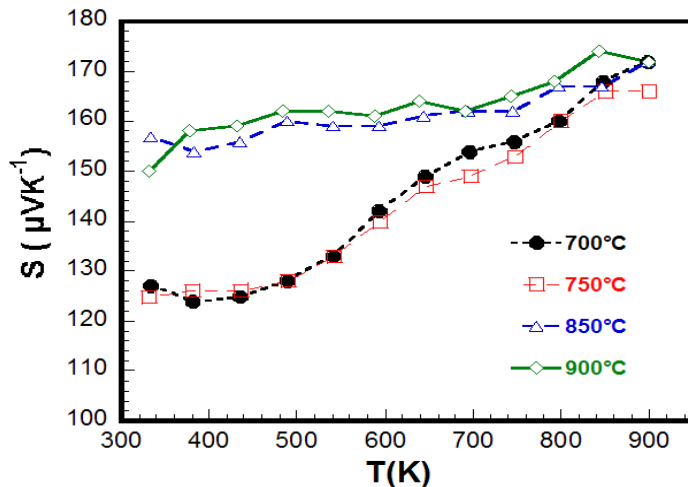
➤ Resistivity and Seebeck coefficient measurements



□ ρ reduces with the SPS temperature

- due to the enhancement in the bulk density

□ Most reduced ρ values are obtained for 900°C

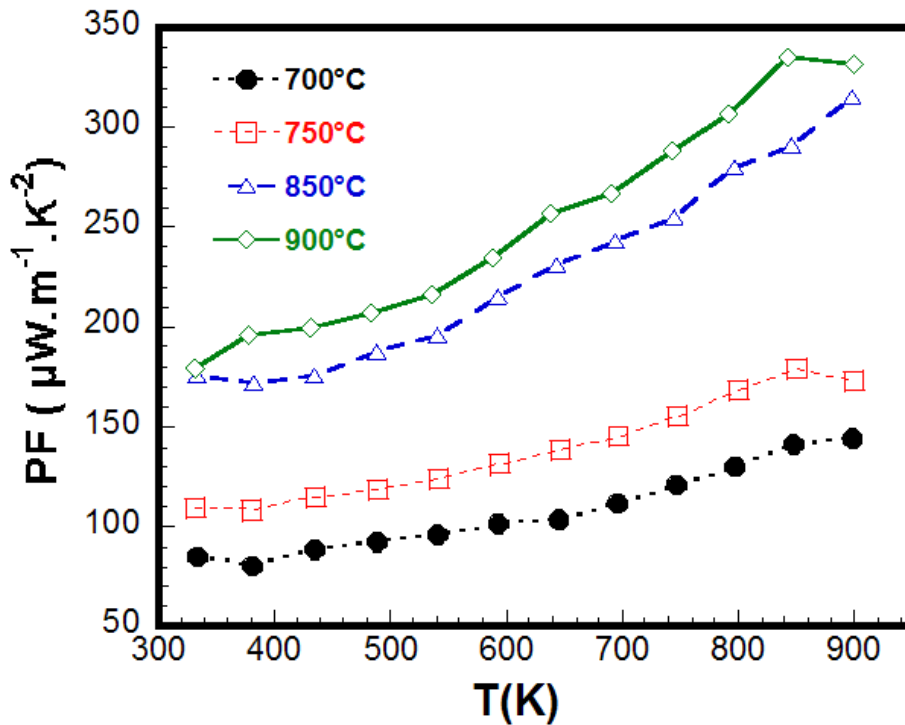


□ Seebeck coefficient values are lower for temperatures of 700 and 750°C compared to those of 850 and 900°C.

- Due to the difference in chemical composition

Thermoelectric characterization

➤ Power factor, $PF = \frac{S^2}{\rho}$



□ PF improves with the dwell temperature

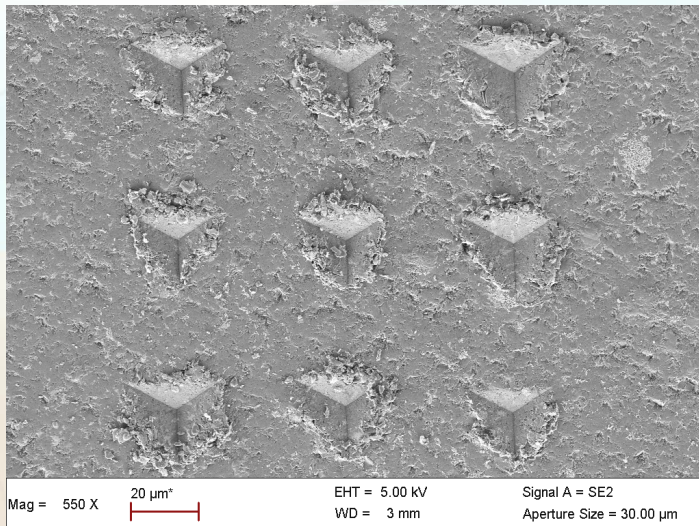
□ PF reaches **315 $\mu\text{W}\cdot\text{m}^{-1}\cdot\text{K}^{-2}$** at **840 K** for a sample processed at **900°C**. [Y. Zhou et al – 2003]
[Y. Liu et al. 2005]

➔ **Better TE properties are obtained for 50 MPa and 900°C**

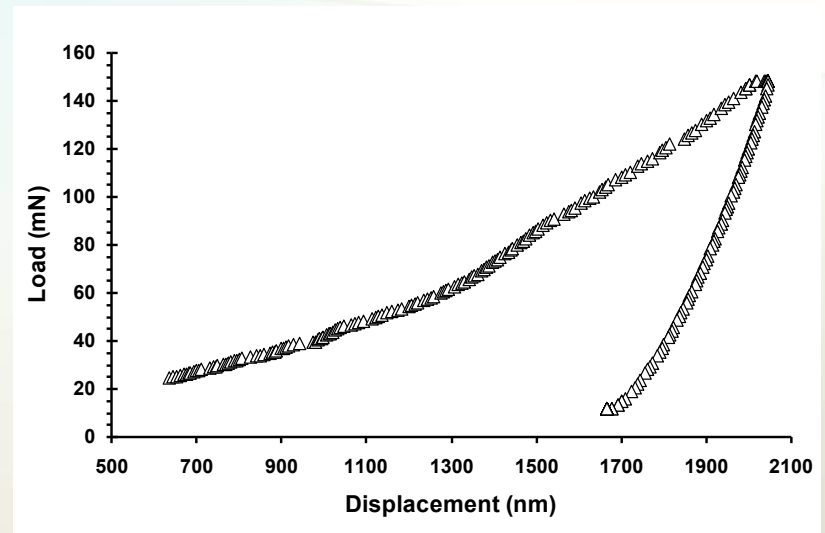
Mechanical properties

■ Nanoindentation (Indenter XP MTS)

■ Principle



Pattern of residual impressions



Load-displacement curve

Mechanical properties

■ Young's modulus, E

$$E = (1-\nu^2) [1/E_r - (1-\nu_i^2) / E_i]^{-1}$$

$E_r = (\pi/2.S)/(2.\beta.A^{1/2})$: the reduced modulus

$S = dP/dh$: the elastic stiffness; $\beta = 1.0615$

ν and ν_i are the Poisson's ratios of the specimen and indenter, respectively.

E_i : the elastic modulus of the indenter.

[W.C. Oliver – 1992]

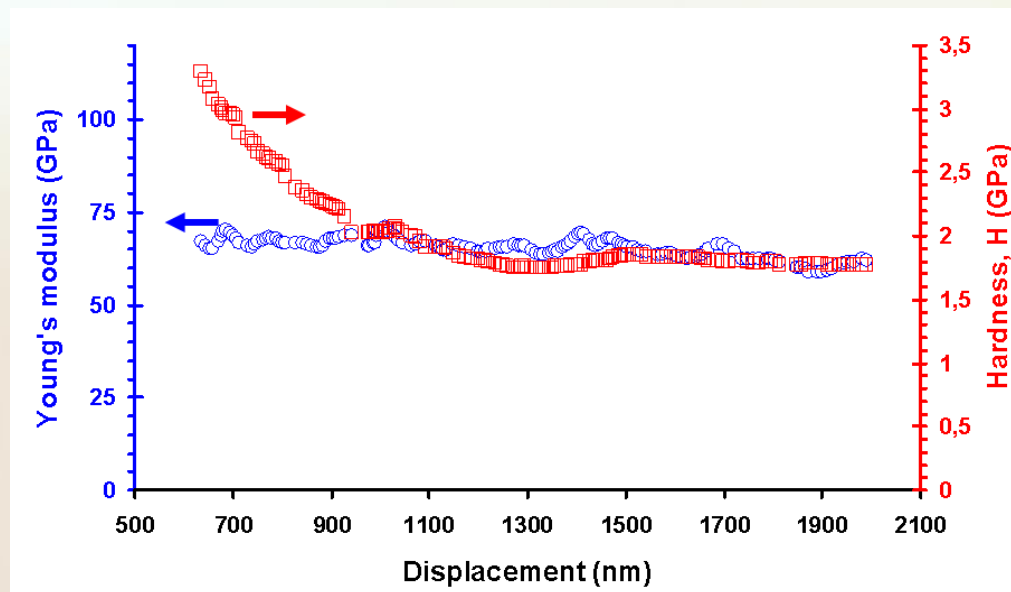
[J. Woirgard – 1997]

■ Hardness, H

$$H = P_{max}/A$$

P_{max} : the load at the maximum penetration depth

A : the projected contact area between the indenter and the sample at P_{max} .

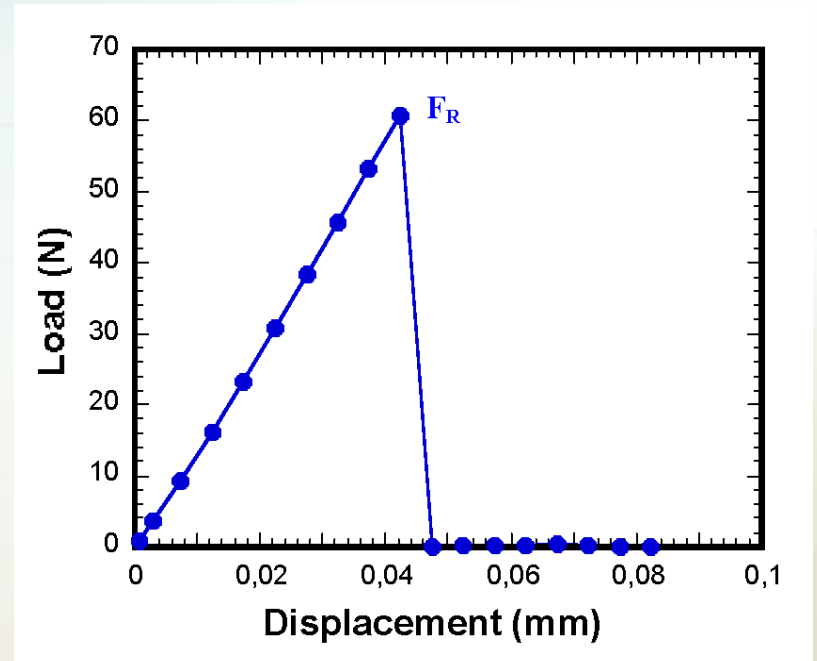
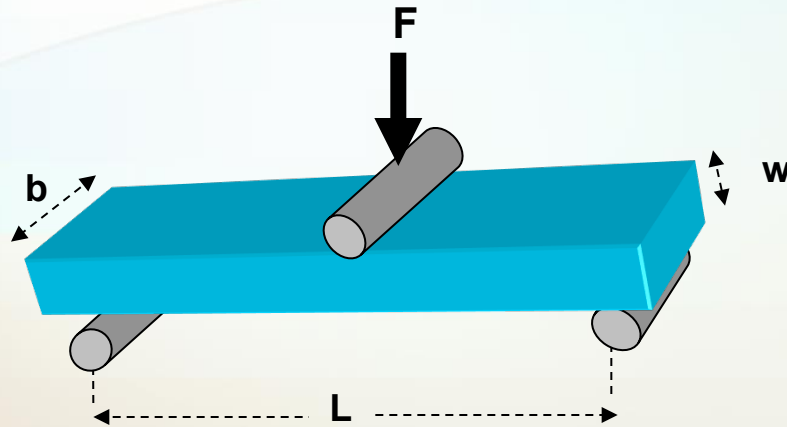


Hardness and Young's modulus versus displacement

Mechanical properties

■ Three point bending test

➤ The strength, σ_R



Load vs displacement for the sample processed under 50 MPa and at 900°C.

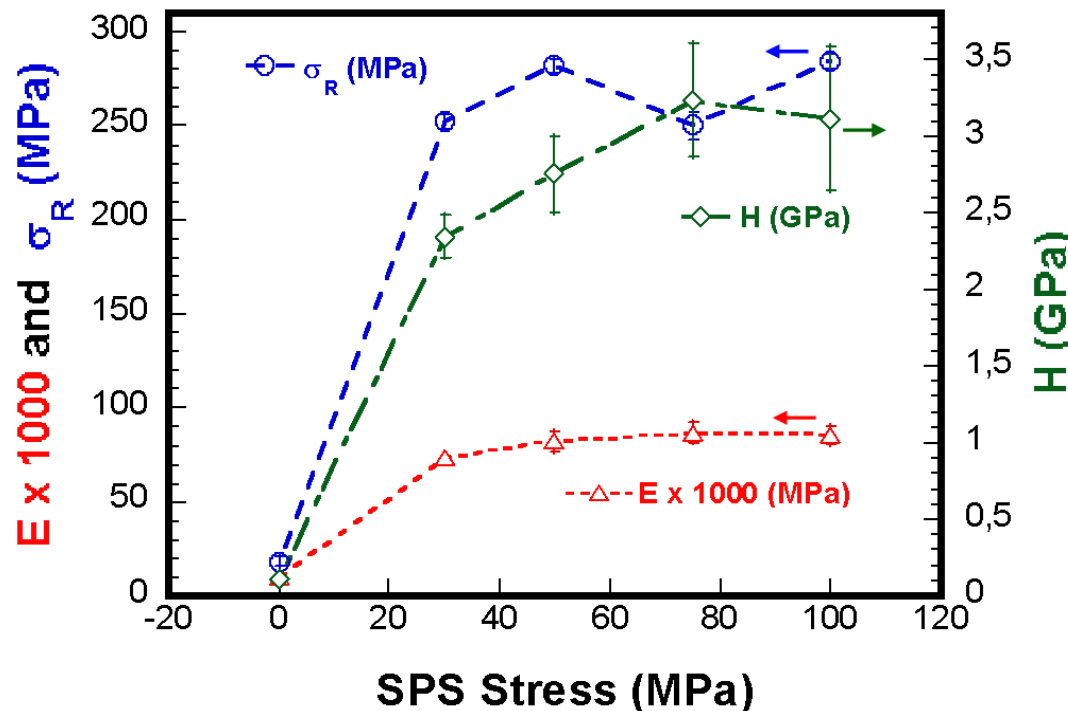
$$\sigma_R = \frac{3 F_R L}{2 b w^2}$$

(F_R : Fracture load)

Mechanical properties

■ Pressing stress effects (0 - 100 MPa)

➤ Hardness (H), Young Modulus (E), and Strength (σ_R), versus SPS pressing stress



□ Considerable improvement of H, E and σ_R with the applied stress.

→ Due to the enhancement in the bulk density

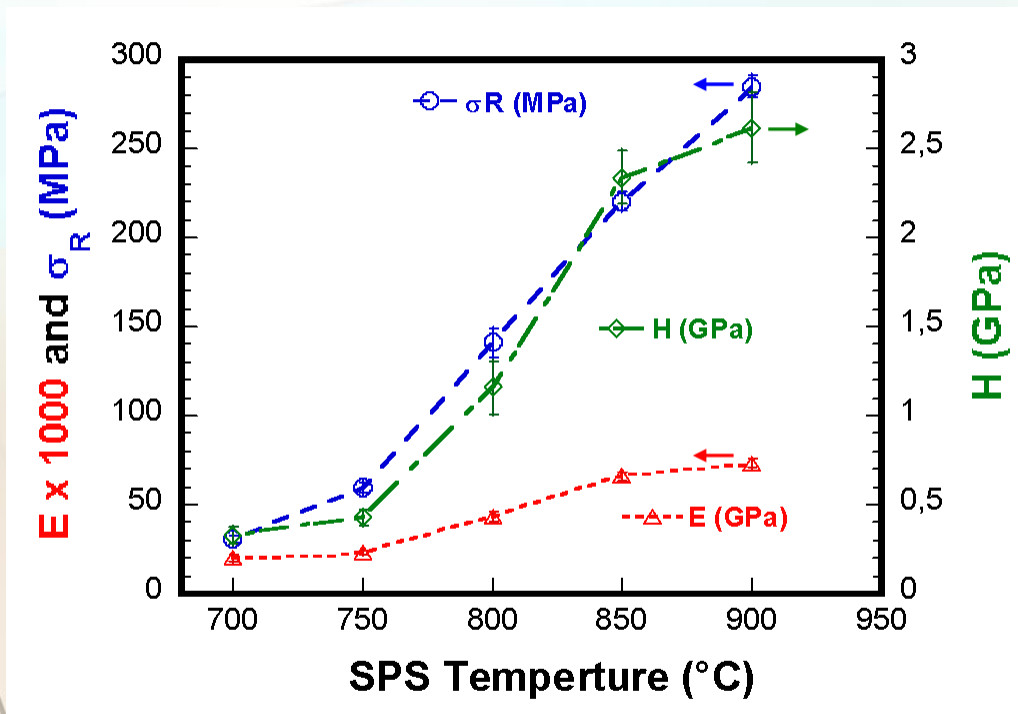
□ A slower increase in H, E and σ_R at larger applied stress

→ Above **50 MPa**, H, E and σ_R are improved more than **29**, **8**, and **15** times, respectively, compared to the reference sample.

Mechanical properties

■ Temperature effects (700 – 900°C)

➤ Hardness, H, Young Modulus, E, and Strength, σ_R , versus SPS temperature.



□ Improvement of H, E and σ_R with the SPS temperature.

Due to :

- the enhancement in the bulk density,
- The chemical composition ??

→ Highest H, E and σ_R values are obtained for **900°C**.

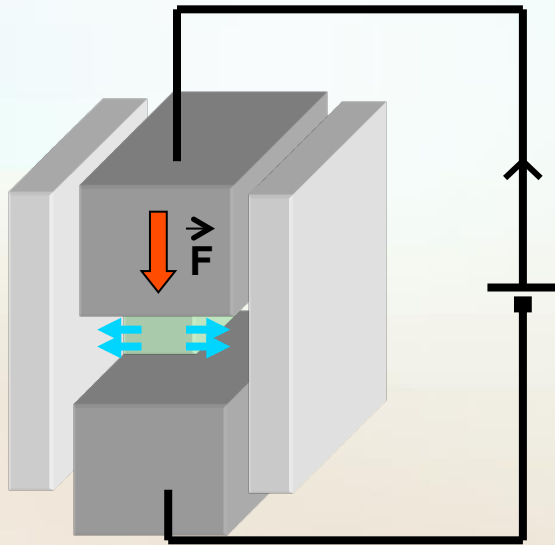
→ At **900°C**, H, E and σ_R recorded an enhancement of **8**, **3.5** and **9** times, respectively, compared to **700°C**.

Conclusions

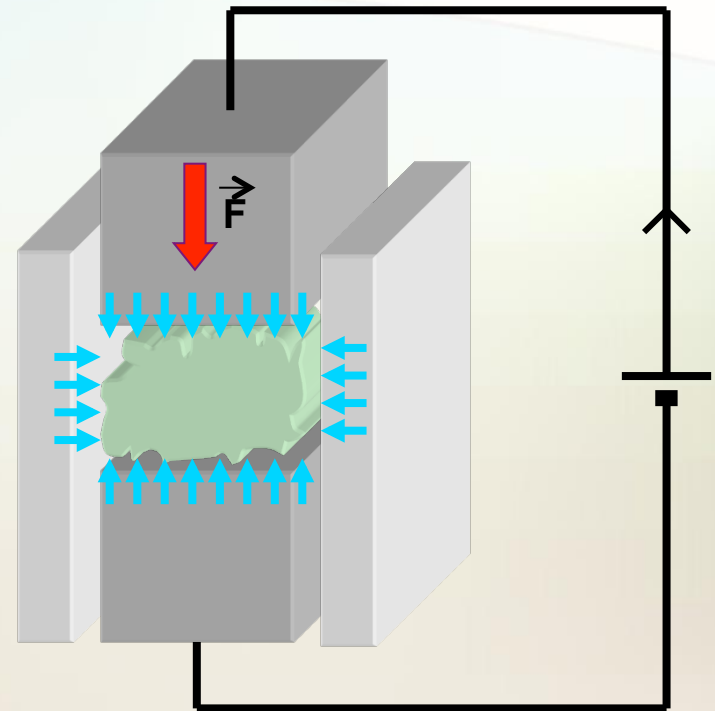
- The **SPS** process was used to consolidated the 349 TE ceramics
- ➔ The optimal thermomechanical cycle is :
50 MPa / 900°C / 2 min
- ➔ Rapid elaboration of 349 ceramics highly densified (**99.7%**)
- ➔ Enhancement of the **thermoelectric performances**
 - PF at 840 K is increased from **70** (reference sample) to **315 $\mu\text{W}\cdot\text{m}^{-1}\cdot\text{K}^{-2}$**
- ➔ Strong improvement of the **mechanical properties**
 - H, E and σ_R are improved more than **29**, **8** and **15** times, respectively, compared to the reference sample

Current and future work

- Use new configurations at **900°C** and **50 MPa** in SPS to allow more texturation, grains size increase and densification



Starting material: Pre-sintered pellet
Free deformation configuration



Starting material: Powder
Conventional SPS

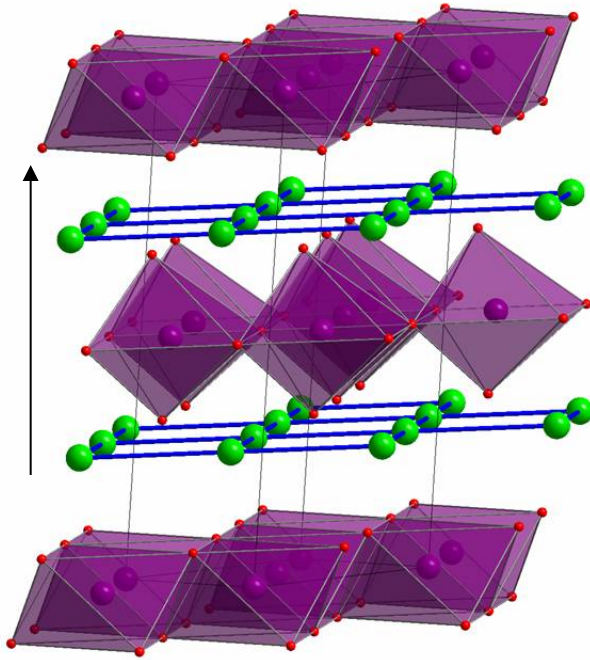
Current and future work

- Fracture toughness tests
 - Thermal conductivity measurements → ZT
- ➔ Integration of 349 ceramics in thermoelectric modules



[S. LEMONNIER – 2008]

Thank you for your attention!



$\text{CoO}_2 = \text{S}_2$ (type CdI_2) $S1: a1=4,838$

Responsible for conduction

$b1=4,557$

$\left. \begin{array}{l} \text{CaO} \\ \text{CoO} \\ \text{CaO} \end{array} \right\} \text{Ca}_2\text{CoO}_3 = \text{S}_1$ $S2: a2=a1$

(type $\text{NaCl} = \text{RS}$) $b2=2,819,$

$\left. \begin{array}{l} \text{CaO} \\ \text{CoO} \\ \text{CaO} \end{array} \right\}$

$c2=c1$

A.C. Masset et al., Phys. Rev. B, 62-2000

$\text{Ca}_3\text{Co}_4\text{O}_9$: Misfit-layer structure with hexagonal CoO_2 layers, related to the CdI_2 type, and square Ca_2CoO_3 layers related to the NaCl type.

The Ca_2CoO_3 layers between the CoO_2 layers result in highly distorted interface and b-axis lattice misfit.

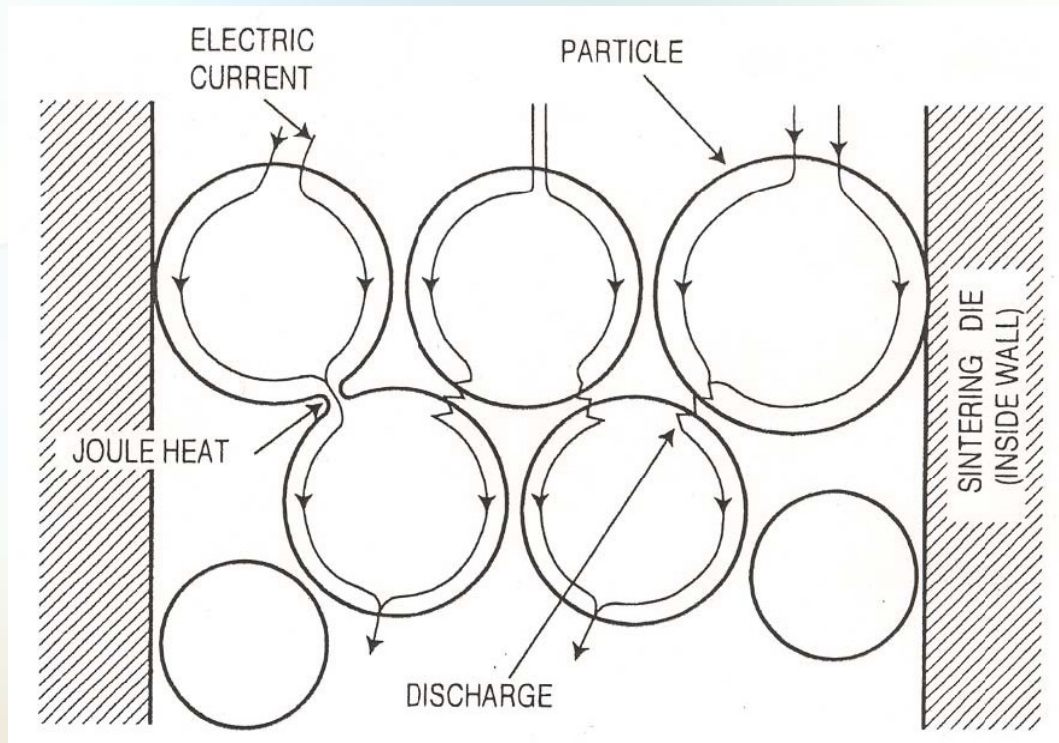
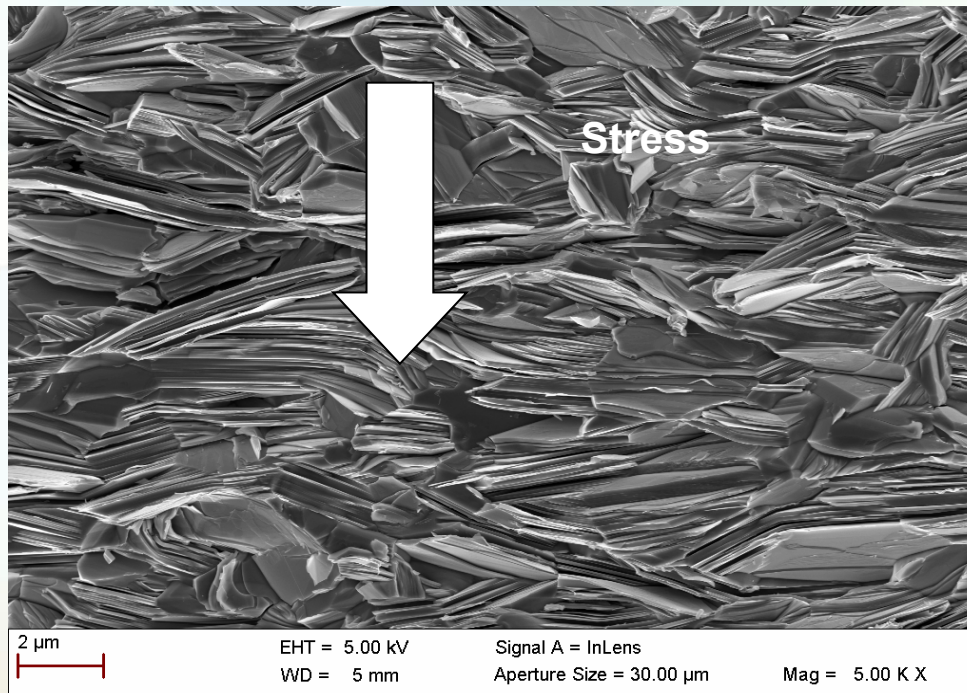


Illustration of how pulse current flows through powder particles inside the SPS sintering die



**SPS-pellet,
900°C/2' /50 MPa**